

## Research Article

# Growth of Cu<sub>2</sub>O Nanopyramids by Ion Beam Sputter Deposition

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In this study, we have successfully deposited n-type Cu<sub>2</sub>O triangular nanopyramids on Si by employing ion beam sputter deposition with an Ar:O<sub>2</sub> ratio of 9:1 at a substrate temperature of 450°C. Scanning electron microscopy measurements showed attractively triangular nanopyramids of ~500 nm edge and height lengths. Both X-ray diffraction and Raman spectroscopy characterizations showed the structures were single-phase polycrystalline Cu<sub>2</sub>O, and the room-temperature photoluminescence investigation showed interestingly green and blue exciton luminescence emissions. All Mott–Schottky, linear sweep voltammetry, and photocurrent measurements indicated that the conductivity of the Cu<sub>2</sub>O pyramids is of n-type.

## 1. Introduction

Recently, cuprous oxide (Cu<sub>2</sub>O) nanostructures are drawing great attention due to their smart optoelectronic characteristics such as nontoxic, naturally abundant, inexpensive market price, and high absorption coefficient. Cu<sub>2</sub>O nanostructures have extensive applications in water splitting [1], photosensing [2], electrode material for lithium-ion batteries [3], photocatalysis [4], low-cost solar energy conversion [5], etc. On the other hand, the use of Cu<sub>2</sub>O as an initial study material to understand the basic principles is indispensable since excitons in Cu<sub>2</sub>O were suggested as noble candidates for understanding of Bose–Einstein condensation [6] because of their unique combinational properties. However, there are few reports on room-temperature exciton emission due to optical quenching and domination of copper vacancy ( $V_{Cu}$ ) related luminescence. As a result, there is no sufficient information about room temperature (RT) excitons to explore the optoelectronic applications of Cu<sub>2</sub>O comprehensively. To overcome these problems, one possible way is to deposit Cu<sub>2</sub>O with suppressed copper vacancies, and this leads to the realization of n-type Cu<sub>2</sub>O since p-type conductivity is due to the existence of copper vacancies.

So far, different methods such as chemical bath deposition, solution-phase epitaxial growth, magnetron sputtering, and electrodeposition [7–10] have been employed to deposit Cu<sub>2</sub>O

nanostructures with different shapes such as nanowires, nanorods, nanocubes, nanopyramids, and polyhedrons. However, most of the methods utilize different precursors and surfactants for control synthesis with relatively high oxygen flow rates. As a result, most of the methods yield p-type Cu<sub>2</sub>O. There are only few reports on the preparation of n-type Cu<sub>2</sub>O nanostructures using physical methods with suppressed copper defect-related luminescence. In this paper, we report the growth and characterization of novel n-type Cu<sub>2</sub>O triangular nanopyramids (TNPs) using ion beam sputter deposition (IBSD), and the experimental results show that attractively high structural quality n-type Cu<sub>2</sub>O TNPs can be fabricated that exhibit RT green exciton photoluminescence (PL).

## 2. Experimental

A copper target was placed at 35 mm downstream of the ion source and a Si substrate was placed at 65 mm upstream of the copper target. Argon and oxygen gases were used as sputter and reactive gases [11], respectively with a flow rate of 9:1. The deposition was performed applying a discharging voltage of 1 kV at a temperature of 450°C for 1.5 hours. Field emission scanning electron microscopy (FE-SEM, JEOL JSM-6500F, and 15 keV) was used to take images. X-ray diffraction (XRD) was measured with a Bruker, D2 Phaser X-ray diffractometer using the Cu K $\alpha$ , radiation ( $\lambda = 0.15406$  nm) in the  $\theta$ – $2\theta$

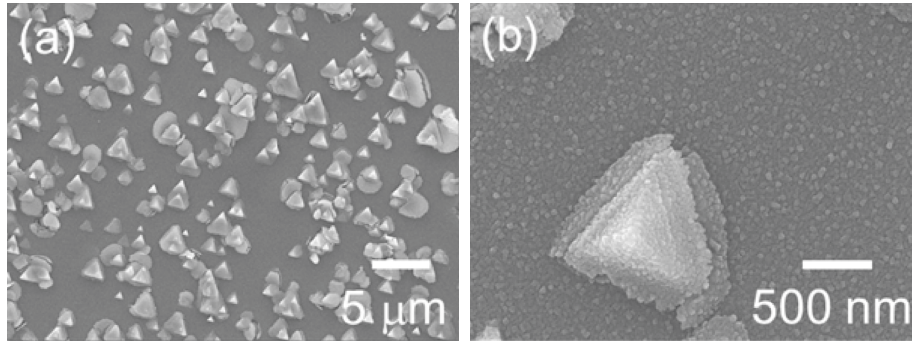


FIGURE 1: FE-SEM micrographs of  $\text{Cu}_2\text{O}$  TNPs.

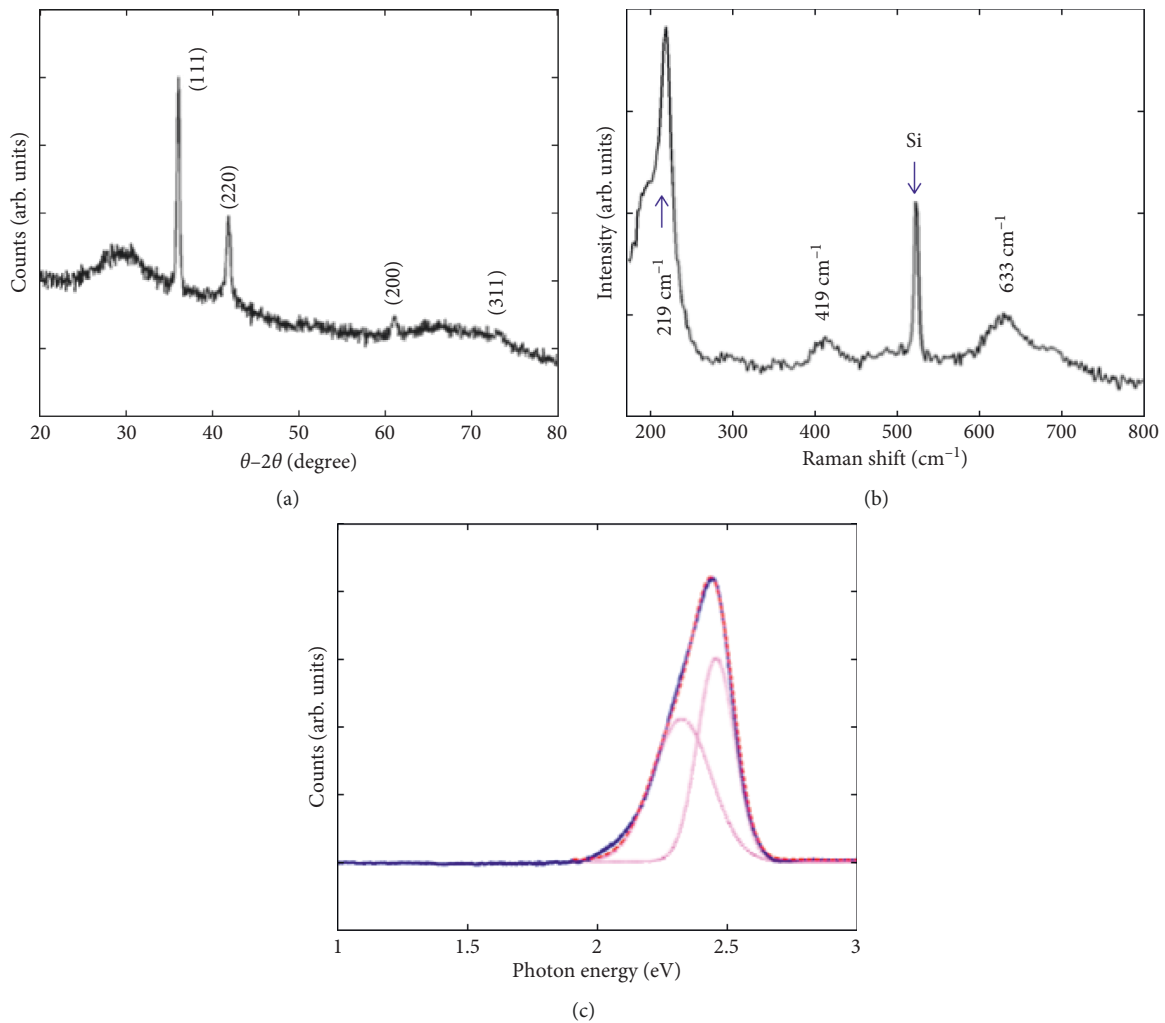


FIGURE 2: (a) XRD diffraction patterns (b) Raman spectra of  $\text{Cu}_2\text{O}$  TNPs and (c) PL spectra of  $\text{Cu}_2\text{O}$  TNPs.

range. Raman spectroscopy measurements were taken in a PTT RAMaker micro-Raman system utilizing a green laser at 532 nm with a power of 10 mW. The RT PL measurement was taken at 300 K using a 405 nm wavelength laser source with a power of 5 mW. The spectra were dispersed by a Triax 550 spectrometer and detected by a CCD detector cooled to  $-71$  K. Photo-electrochemical measurements were done by a Gamry G300 potentiostat with  $\text{Ag}/\text{AgCl}$ , Pt, and  $\text{Cu}_2\text{O}$  sample

as reference, counter, and working electrodes, respectively, using a customized electrochemical cell filled with 0.5 M  $\text{K}_2\text{SO}_4$  electrolyte, employing xenon lamp for illumination.

### 3. Results and Discussion

Figures 1(a) and 1(b) denote the FE-SEM micrographs of  $\text{Cu}_2\text{O}$  TNP samples with different magnification values. The

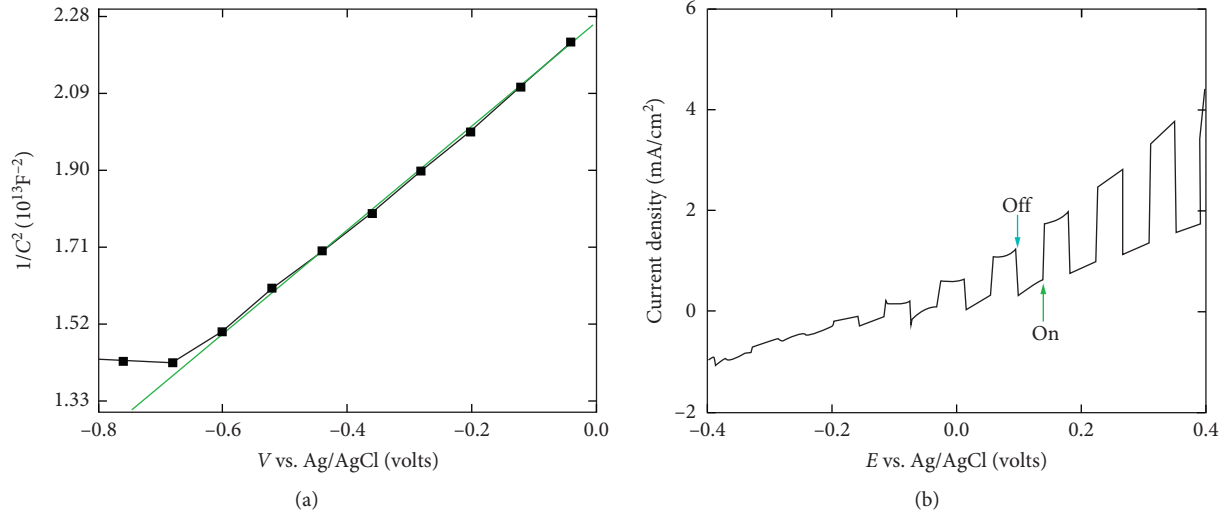


FIGURE 3: (a) Mott-Schottky plot and (b) linear sweep voltammetry measurement of  $\text{Cu}_2\text{O}$  TNPs.

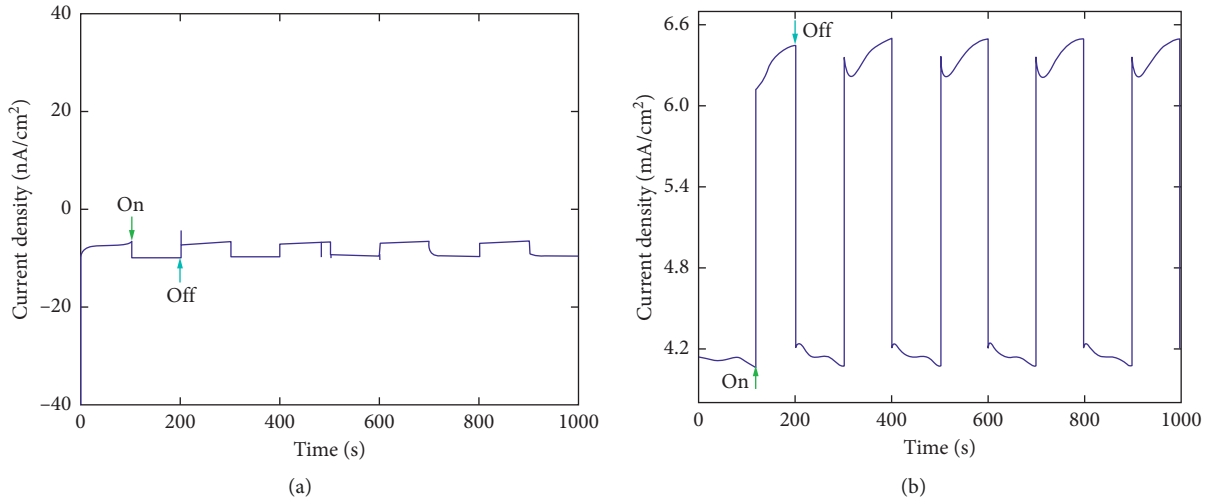


FIGURE 4: (a)  $-0.3$  V and (b)  $0.3$  V biased transient photocurrent graphs for  $\text{Cu}_2\text{O}$  TNPs.

images indicate uniform, vertically aligned  $\text{Cu}_2\text{O}$  TNPs with an edge length of  $\sim 500$  nm, and the images exhibit smooth morphologies. The cross-sectional view of the FE-SEM micrograph (not presented) shows that the  $\text{Cu}_2\text{O}$  TNPs are on the top of a  $\sim 180$  nm thick  $\text{Cu}_2\text{O}$  thin film, and the height of the TNPs is  $\sim 500$  nm. This demonstrates that the growth mechanism of the  $\text{Cu}_2\text{O}$  TNPs is due to the Stranski-Krastanov (SK) mode of growth.

Figure 2(a) demonstrates the XRD patterns of the  $\text{Cu}_2\text{O}$  TNP samples, and it shows all the diffraction peaks matched with  $\text{Cu}_2\text{O}$  phase (JCPDS #78-2076) and are indexed as the (111), (200), (220), and (311) planes. As it can be seen from Figure 2(a), the (111) peak is the largest compared with the other peaks, and it shows the growth of the  $\text{Cu}_2\text{O}$  TNPs is along the (111) plane. The sample was further investigated using Raman spectroscopy (Figure 2(b)). The measurement shows all the peaks are the characteristic Raman peaks of  $\text{Cu}_2\text{O}$  [12–15], and the result is consistent with the XRD

measurements. Moreover, the strongly defined Raman mode at  $219 \text{ cm}^{-1}$  proves the structural quality of the sample.

Photoluminescence (PL) characterization of  $\text{Cu}_2\text{O}$  TNPs was carried out at room temperature. Figure 2(c) demonstrates the PL response of the  $\text{Cu}_2\text{O}$  TNP sample at RT, and the result shows a strong PL exciton emission of  $2.451 \text{ eV}$ . This strong PL band can be fitted into two Gaussian peaks: the green exciton ( $2.3 \text{ eV}$ ) and blue exciton ( $2.451 \text{ eV}$ ) [16] of  $\text{Cu}_2\text{O}$ . Mostly, exciton emission of  $\text{Cu}_2\text{O}$  is observed at low temperature ( $\sim 5 \text{ K}$ ), but this study clearly shows a promising RT exciton emission. According to Ito et al., the green exciton emission of cuprous oxide can be observed at  $2.304 \text{ eV}$  at a temperature of  $4.2 \text{ K}$ , which is similar to our result ( $2.3 \text{ eV}$ ) attributed due to transition from  $\Gamma_8^+$  to  $\Gamma_6^+$ . Furthermore, the PL peak at  $2.451 \text{ eV}$  is closer to the blue luminescence emission  $E_{\text{oc}}$  ( $2.6 \text{ eV}$ ) observed during the transition from  $\Gamma_8^+$  to  $\Gamma_8^-$ . The detection of these significantly strong RT exciton emissions from the sample is a very

encouraging result, and it has extensive potential applications in fabrication and engineering of photonic and optoelectronic devices working in the green and blue light spectrum.

As shown from Figure 3(a), the value of slope of the Mott-Schottky plot of the Cu<sub>2</sub>O TNPs is positive, demonstrating a typical behavior of n-type conductivity with a linear increase of  $C^{-2}$  when the voltage becomes larger and larger. Figure 3(b) shows the linear sweep voltammetry measurement of the Cu<sub>2</sub>O TNP sample carried out in a custom-built system, which includes a light source, an illumination switch, and a three-electrode cell system to investigate the conduction type of the sample recorded in 0.5 M of K<sub>2</sub>SO<sub>4</sub> electrolyte. Figure 3(b) shows that as the potential becomes negative, the anodic photocurrent drops. On the other hand, as the voltage becomes positive, the anodic photocurrent increases, which proves the nature of n-type photoelectrodes.

Figure 4 designates the photocurrent response of Cu<sub>2</sub>O TNPs under  $-0.3$  V (Figure 4(a)) and  $0.3$  V (Figure 4(b)) bias with respect to the Ag/AgCl reference electrode. The figures show anodic photocurrent appeared dominantly even though there is negligible cathodic photocurrent density (Figure 4(a)) when biased negatively compared with the positively biased. The magnitude of the anodic photocurrent for the positively biased (Figure 4(b)) one is very huge, indicating that the carrier type is of n-type, and the value of the photocurrent density ( $\sim 2.1$  mA/cm<sup>2</sup>) is very huge and promising for the Cu<sub>2</sub>O semiconductor material.

#### 4. Conclusion

In this work, n-type Cu<sub>2</sub>O TNPs with high structural quality and excellent optical properties have been grown successfully by IBSD with an Ar : O<sub>2</sub> ratio of 9 : 1 at a temperature of 450°C using metallic copper as a target. The FE-SEM investigation shows that Cu<sub>2</sub>O TNPs of edge length  $\sim 500$  nm have been grown across the sample on the top of the Cu<sub>2</sub>O thin film (TF), following the SK mode of growth. Both XRD and Raman spectra measurements reveal with good agreement that the sample is a single-phase Cu<sub>2</sub>O. Strong exciton PL bands observed at 2.3 eV and 2.45 eV attributed from green exciton and blue PL emissions. In this study, the observation of the exciton luminescence in the green and blue regions of the spectrum of light is very useful for further understanding of the optical properties of Cu<sub>2</sub>O nanostructures and for the fabrication of optoelectronic displaying devices working at RT using Cu<sub>2</sub>O TNPs. Besides, the high structural quality and the n-type conductivity of the sample ascertain that the grown pyramids are of novel quality and can be used in low-dimensional semiconductor researches and helpful to improve the efficiency of the solar cell using Cu<sub>2</sub>O.

#### Data Availability

The data used to support the findings of this study are available from the corresponding author upon request.

#### Conflicts of Interest

The author declares that there are no conflicts of interest regarding the publication of this paper.

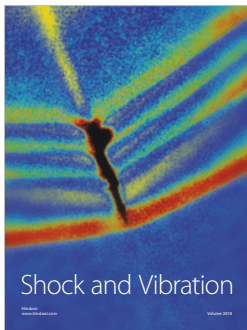
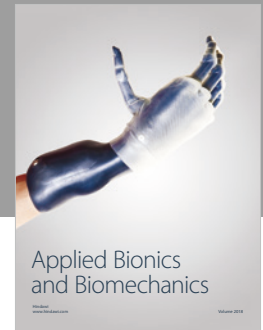
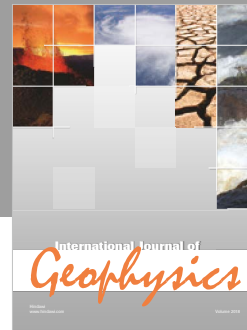
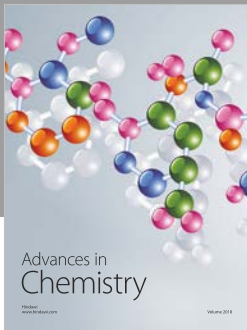
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