Hindawi Publishing Corporation International Journal of Photoenergy Volume 2015, Article ID 261372, 7 pages http://dx.doi.org/10.1155/2015/261372



Research Article

Hydrothermal Growth of Quasi-Monocrystal ZnO Thin Films and Their Application in Ultraviolet Photodetectors

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Received 28 October 2014; Revised 31 January 2015; Accepted 2 February 2015

Academic Editor: Shyh-Jer Huang

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Quasi-monocrystal ZnO film grown using the hydrothermal growth method is used for the fabrication of Cu_2O/ZnO heterojunction (HJ) ultraviolet photodetectors (UV-PDs). The HJ was formed via the sputtering deposition of p-type Cu_2O onto hydrothermally grown ZnO film (HTG-ZnO-film). The effect of annealing temperature in the nitrogen ambient on the photoluminescence spectra of the synthesized ZnO film was studied. The optoelectronic properties of Cu_2O/ZnO film with various Cu_2O thicknesses (250–750 nm) under UV light (365 nm; intensity: 3 mW/cm²) were determined. The UV sensitivity of the HTG-ZnO-film-based UV-PDs and the sputtered ZnO-film-based UV-PDs were 55.6-fold (S_{HTG}) and 8.8-fold ($S_{sputter}$), respectively. The significant gain in sensitivity ($S_{HTG}/S_{sputter} = 630\%$) of the proposed ZnO-film-based device compared to that for the device based on sputtered film can be attributed to the improved photoelectric properties of quasi-monocrystal ZnO film.

1. Introduction

Ultraviolet photodetectors (UV-PDs) have attracted considerable interest for applications such as combustion engineering, missile plume detection, space-to-space communication, ozone layer monitoring, pollution monitoring, and flame detection [1, 2]. UV-PDs have been fabricated on wide-direct-band-gap materials, including ZnO film obtained via spray deposition [3], GaN film obtained via plasma-assisted molecular beam epitaxy [4], and TiO₂ obtained via hydrothermal growth (HTG) [5]. Among them, ZnO is a well-known direct wide band gap (3.37 eV at room temperature) and thermally stable n-type wurtzite semiconductor with a large exciton binding energy (about 60 meV) [6]. It has been widely investigated for its unique properties and potential applications in sensitive devices [7–15].

ZnO-based UV-PDs with a nanowire- (NW) type structure have drawn a lot of attention due to their ease of fabrication, low-temperature processing, and unique properties, such as high aspect ratio, high surface-to-volume ratio,

and good carrier confinement in two dimensions, which benefit device performance [16]. However, UV-PDs based on ZnO NWs usually suffer from a relatively low junction area for photo detection. In addition, the integration of ZnO NWs into a working device remains a complicated, time-consuming, and uneconomical process, and there is a potential short-circuit problem for ZnO-NW-based UV-PDs [17, 18].

In this study, the fabrication of UV-PDs based on a p-Cu₂O/n-ZnO heterojunction (HJ) is demonstrated. A quasimonocrystalline ZnO film grown on an indium tin oxide-(ITO) coated glass substrate via the hydrothermal method is proposed to serve as the n-type semiconductor with improved photoelectric properties. Sputter deposition is used to deposit cupric oxide (Cu₂O) as the p-type semiconductor. In addition to nontoxicity, low cost, and abundance of the starting material, Cu₂O has a direct band gap of 2.0 eV [19, 20]. A sensing response with a wide spectrum range of 660~ 300 nm can be expected for the proposed p-Cu₂O/n-ZnO HJ. The effects of Cu₂O film thickness on the optoelectrical

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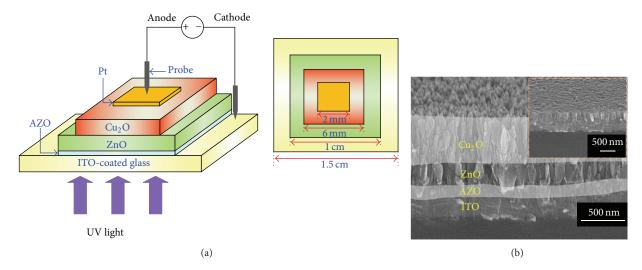


FIGURE 1: Schematic device structure and SEM images of prepared UV-PDs. (a) Schematic device structure and (b) tilted-view images of ZnO (HTG)/AZO/ITO and p- Cu_2O/ZnO (HTG)/AZO/ITO structures.

properties of UV-PDs are investigated. Comparisons between the performances of p-Cu₂O/n-ZnO UV-PDs based on n-type ZnO films prepared via HTG and sputtering deposition, respectively, are made.

2. Experimental Details

Figure 1(a) shows the device structures of ZnO-based UV-PDs presented in this work. First, ITO-coated glass substrates were cleaned in a sonicating bath of acetone and isopropanol for 30 min, rinsed in deionized water for 10 min, and dried with nitrogen gas. The ITO-coated glass substrates were used for the fabrication of an n-side-up vertical-structured PD devices. Essentially, the ITO glass substrate is with a high electrical conductivity and high optical transparency ITO coating layer atop glass substrate, which could serve as an excellent transparent electrode for optoelectronic devices with bottomup fabrication process. In addition, it offers a much shorter length for forward current path (~1 \mu m), better current uniformity, and smaller series resistance, as compared with regular PD devices fabricated on glass substrate or insulated substrate. A 150 nm thick aluminum-doped zinc oxide (AZO) film was sputtered on ITO-coated glass substrates using radiofrequency (RF) sputtering with an RF power of 60 W for 90 min to serve as a seed layer. These samples were placed in a solution of 0.06 M zinc nitrate hexahydrate $(Zn(NO_3)_2 \cdot 6H_2O)$ and hexamethylenetetramine $(C_6H_{12}N_4)$ at 70°C for 4h for the growth of ZnO film (300 nm). At the end of the reaction, the samples were removed from the solution, washed with deionized water for 5 min, and dried in nitrogen gas. Subsequently, p-Cu₂O films with three thicknesses (250, 500, and 750 nm) were sputtered onto the surface of the ZnO film by RF sputtering with a power of 90 W for 3, 6, and 9 h, respectively, to complete the fabrication of p-Cu₂O/n-ZnO HJs. To examine the impact of annealing temperature on material quality, the samples were annealed in N_2 gas at 300, 400, 500, and 600°C for 10 min, respectively.

Finally, an ohmic-contact electrode with 200 nm thick Pt film electrodes was deposited on the surface of the p-Cu₂O layer via e-gun evaporation. For comparison, p-Cu₂O (500 nm)/n-ZnO (300 nm) UV-PDs based on sputter-deposited ZnO film of the same thickness and annealed in N₂ gas at 600°C for 10 min were also fabricated. All UV-PDs reported in this work have a die size of 1.5 cm \times 1.5 cm and a working area of 6 mm \times 6 mm

The surface morphologies and thicknesses of the grown ZnO film and p-Cu₂O/n-ZnO structure were characterized using scanning electron microscopy (SEM; Hitachi SU8000) at an accelerating voltage of 10 keV. The crystal structure and optical properties of ZnO films fabricated via HTG were examined using X-ray diffraction (XRD; D/MAX2500, Rigaku) with $CuK\alpha$ (wavelength: 1.5418 Å) radiation at a scan step of 0.01° (2 θ), UV-visible-near-infrared double-beam spectrophotometry (U-3010 UV-VIS) with a 300-900 nm spectral range, and photoluminescence (PL) with an He-Cd laser as the light source using an excitation wavelength of 325 nm and a power of 3 mW, respectively. The current density-voltage (J-V) characteristics of the p-Cu₂O/n-ZnO film fabricated via HTG were measured using a Keithley 2636 voltage source instrument at room temperature in darkness and under 500-W Xe arc lamp illumination.

3. Results and Discussion

SEM images of the grown ZnO film and the prepared p-Cu₂O (500 nm)/n-ZnO structure on an AZO/ITO substrate are shown in Figure 1(b). ZnO film with a controllable thickness (300 nm) was obtained by regulating the solution concentration, temperature, and growth time. As revealed by the inset in the figure, the grown ZnO film fully covered the AZO/ITO substrate, which effectively prevented the p-type material (Cu₂O) from reaching the substrate during HJ formation and avoided the possible short-circuit problem usually encountered for ZnO-NW-based UV-PDs [17, 18].

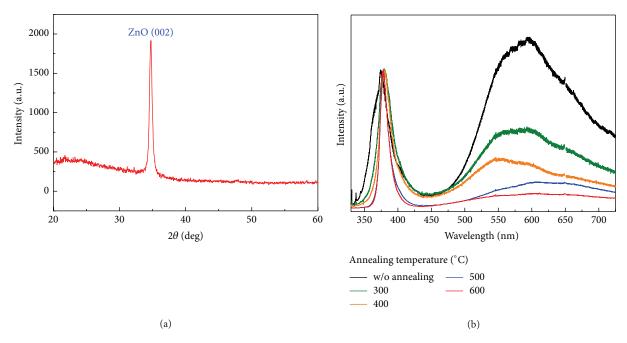


FIGURE 2: (a) XRD pattern of prepared n-ZnO film with quasi-monocrystal structure fabricated via HTG. (b) PL spectra of ZnO film (HTG) for various annealing temperatures in nitrogen.

The mechanism of the growth of the quasi-monocrystal ZnO film on the ITO-coated glass substrate can be attributed to the well-controlled parameters, such as solution concentration, temperature, and growth time, of the HTG process. It has been reported that these parameters have a strong impact on the aspect ratio of ZnO NWs [21]. A lower HTG temperature, higher solution concentration, and longer growth time lead to a larger diameter of ZnO NWs. ZnO films were formed through the aggregation of NWs and then followed by self-organized growth among interconnected NWs [22]. As shown in Figure 1(b), the sputter-deposited p-Cu₂O film had a dense NW-like morphology. It is speculated that the sputtering power used in this work might have enough energy to increase the nucleation kinetics during sputter deposition, which is in good agreement with the experimental findings on sputter-deposited Cu₂O films with various sputtering powers [23, 24].

The material properties of the prepared ZnO film were analyzed by XRD and PL, with the results shown in Figure 2. On the basis of the diffraction peak that corresponded to the (002) plane shown in Figure 2(a), the ZnO film fabricated via HTG was indexed to have a hexagonal wurtzite signal according to the standard JCPDS card [25]. This indicates that ZnO films fabricated via HTG are mainly formed via self-organized growth interconnections of closely packed crystalline columnar ZnO with the c-axis vertical to the substrate surface. It is believed that the ZnO films fabricated via HTG in the present work do have a quasi-monocrystalline structure. The influence of thermal annealing conditions on the PL spectra of the ZnO films fabricated via HTG on a controlled wafer was examined. The results are shown in Figure 2(b). The ZnO films with and without annealing exhibit UV and visible (VIS) emission peaks with different

full widths at half maximum (FWHM) values for the former and different intensities for the latter. The UV and VIS emission peaks are attributed to free exciton recombination and a transition of excited optical centers from a deep level to the valence level arising from singly ionized oxygen vacancies [26], respectively. Our results show that the intensity of the VIS light emission peak decreases with increasing annealing temperature. This might be due to the nitrogen atmosphere suppressing the reevaporation of oxygen during annealing. At the same time, nonradiative-related defects may have been reduced, increasing UV emission [27]. As a result, a considerable decay in VIS emission intensity of the nitrogenannealed ZnO films fabricated via HTG was observed.

Figure 3 shows the measured *J-V* curves of the p-Cu₂O/ZnO film (HTG) HJs in darkness. The fabricated HJs with various Cu₂O thicknesses after annealing at 600°C in nitrogen all exhibit well-defined rectifying behaviors. In this work, the electron concentrations in the annealed ZnO film and the hole concentration in the annealed Cu $_2$ O are $1{\sim}2\times10^{16}~cm^{-3}$ and $2{\sim}3\times10^{15}~cm^{-3}$, respectively, based on Hall measurements. The width of depletion region corresponding to the abovementioned carrier concentrations is 350~500 nm (Cu₂O: 300~400 nm; ZnO: 50~90 nm). The band offsets between ZnO and Cu₂O are $\Delta E_c = 0.9 \,\text{eV}$ and $\Delta E_v =$ 2.17 eV [28], respectively. Note that the sample with 250 nm thick Cu₂O film exhibits the highest forward current, which can be attributed to the thickness of the Cu₂O layer being insufficient, and as a result, both the built-in voltage and the depletion region were limited. Similarly, the decrease in the forward current of the 750 nm thick Cu₂O device might be due to the series resistance of the thicker Cu₂O layer. The annealed sample with 500 nm thick Cu₂O film is the most suitable for UV-PDs because it has the best trade-off between

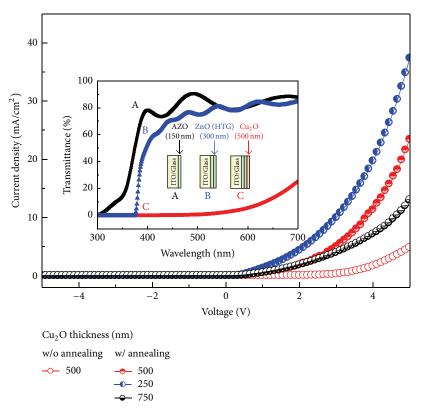


FIGURE 3: Experimental J-V curves of p-Cu₂O/ZnO film (HTG) HJs in darkness. Inset shows the measured transmittance of the prepared samples after annealing. ITO-coated glass substrate after AZO (150 nm) layer deposition (structure A), with an additional ZnO film (HTG) (300 nm) (structure B), and with a Cu₂O layer (500 nm) atop structure B (structure C).

built-in voltage/depletion width and the series resistance. The annealed device with 500 nm thick Cu₂O film exhibits a lower forward threshold voltage ($V_{\rm th}$ at $0.1\,{\rm mA/cm^2}$) of $0.5\,{\rm V}$ and a lower leakage current (J_r) of 9×10^{-8} A/cm² biased at -1 V compared to those of the unannealed devices ($V_{th} = 2.5 \text{ V}$; $J_r = 1.37 \times 10^{-7} \,\mathrm{A/cm^2}$). These results suggest that ZnO films fabricated via HTG have improved material quality, including reduced defect states and enhanced crystallinity, after thermal annealing [29]. The light transmittances of the annealed samples including AZO film (150 nm) and ZnO film (300 nm) fabricated via HTG/AZO and Cu₂O film (500 nm)/ZnO/AZO prepared atop the ITO glass substrate are compared in the inset of Figure 3. The Cu₂O/ZnO/AZO structure has the lowest transmittance (0~1%) in the UV light spectrum, indicating that both Cu₂O and ZnO films have strong UV light absorptivity.

Figure 4 shows the measured reverse-biased *J-V* characteristics of the prepared HJs based on ZnO films fabricated via HTG in the dark or under various illumination conditions. The experimental results indicate that unannealed UV-PDs are responsive to both UV and VIS light and that annealed UV-PDs are only sensitive to UV light, avoiding VIS interference during UV detection. The improved performance of the annealed UV-PDs can be attributed to the suppression of structural and interfacial defects, which decreases the component of photocurrent through defect states to the conduction band in the ambient in the presence of VIS light.

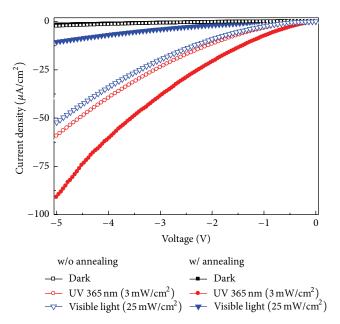


FIGURE 4: Comparisons of experimental *J-V* curves of p-Cu₂O/ZnO (HTG) sample with and without thermal annealing.

Figure 5 shows the spectral response of the annealed UV-PDs measured under a reverse bias of 1 V. It was found that the responsibility shows a cutoff at around 370~390 nm and the

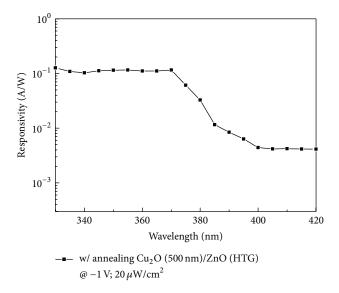


FIGURE 5: Spectral response of the p-Cu₂O/ZnO (HTG) sample UV sensor measured at a reverse bias of 1 V.

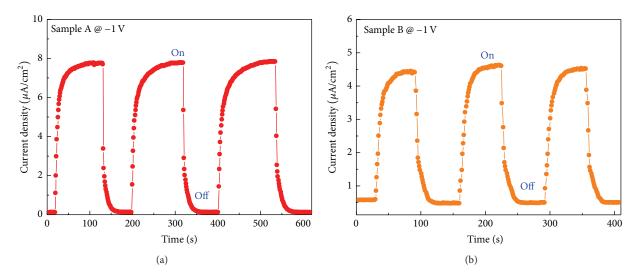


FIGURE 6: Transient response of photocurrent density of fabricated devices based on ZnO film (HTG) (sample A) and sputter-deposited ZnO film (sample B). Wavelength and intensity of UV light are $365 \, \text{nm}$ and $3 \, \text{mW/cm}^2$, respectively.

response of the UV region (330~370 nm) is nearly constant which is higher than that of in the visible spectrum by one order more.

Figure 6 shows the dynamic photoresponses of the UV-PDs annealed at 600°C in nitrogen based on hydrothermally grown and sputter-deposited ZnO films, respectively (referred to as samples A and B, resp.). Note that the thicknesses of the p-Cu₂O and ZnO layers were 500 and 300 nm, respectively. The response times of the devices with sputter-deposited and hydrothermally grown films are about 20 and 22 s and the recovery times are about 26 and 30 s, respectively. The device based on hydrothermally grown film shows a 55.6-fold increase in the current at -1 V with irradiation under 365 nm UV light and a power density of 3 mW/cm² compared to that for the device based on sputter-deposited film. The enhanced photoresponse of the former

device is mainly attributed to the hydrothermal-grown ZnO film which was formed by aggregated single-crystalline ZnO nanowires with the c-axis vertical to the substrate surface. It provides parallel current paths through the coalesced single-crystalline ZnO nanowires with much better optoelectronic properties, as compared with the sputter-deposited ZnO films with polycrystalline or amorphous structure. In addition, thermal annealing offers further improvement in the ZnO film quality through suppressing nonradiative-related defects as revealed from PL measurements.

In recent years, ZnO-based UV PDs have been widely reported. Lee et al. reported the RF sputtering of p-NiO film onto the surface of the ZnO NW by MOCVD and the measured UV sensitivity, response times, and recovery times were 11.3-fold, 72 s, and 110 s, respectively. Wang et al. reported the hydrothermal growth of ZnO NW on CuO

nanowires and the measured UV sensitivity, response times, and recovery times were 3-fold, 100 s, and 141 s, respectively. More recently, Guo et al. presented the hydrothermal growth of ZnO nanorod arrays on NiO nanowires and the measured UV sensitivity, response times, and recovery times were 23-fold, 55 s, and 70 s, respectively [30–32]. In this work, the preparation of p-Cu₂O/ZnO film (HTG) HJs is with measured UV sensitivity, response times, and recovery times which were 55.6-fold, 20 s, and 26 s, respectively, which are comparably much better than those of UV-PDs reported in the literature mentioned above.

4. Conclusion

The preparation of p-Cu₂O/ZnO film (HTG) HJs with good UV sensitivity was demonstrated. ZnO films with a controllable thickness (300 nm) that fully covered the AZO/ITO substrate were obtained by suitably regulating the solution concentration, temperature, and growth time of the HTG process. According to the XRD and PL analyses, the ZnO film fabricated via HTG had a quasi-monocrystalline structure. Thermal annealing in nitrogen at 600°C effectively eliminated defect states in the ZnO films, suppressing VIS interference during UV light detection. Experimental results reveal that the p-Cu₂O/n-ZnO films fabricated via HTG have a fairly good response to UV light (3 mW/cm² at 365 nm) with an increase in the photocurrent of about 55.6-fold compared to that for the device based on sputter-deposited film. It is expected that the HJs based on ZnO film fabricated via HTG will be suitable for future optoelectronic applications.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

Acknowledgments

This work was supported by the Ministry of Science and Technology (MOST) of Taiwan, under Grant MOST 103-2221-E-006-132 and the National Science Council (NSC) of Taiwan, under Grant NSC 102-2221-E-006-217-MY2. The authors would like to thank the National Nano Device Laboratories and the Center for Micro/Nano Science and Technology, National Cheng Kung University, Taiwan, for equipment access and financial and technical support.

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